

# Gate current degradation in W-band InAlN/AlN/GaN HEMTs under Gate Stress

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# Purpose

To understand degradation of gate leakage current  
in ultra-scaled InAlN/AlN/GaN HEMTs  
under positive gate stress

# Outline

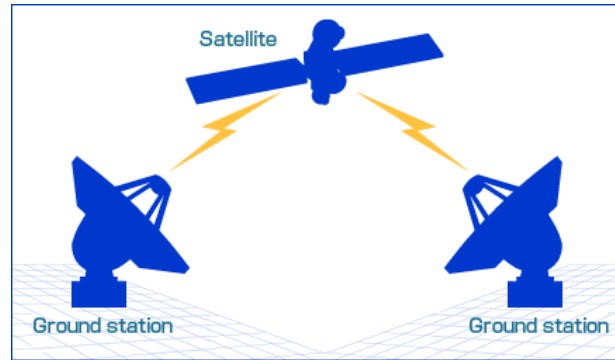
1. Motivation
2. Devices and experimental approach
3. Gate stress experiments
  - Harsh step-stress (-recovery) experiments
  - Mild constant gate stress experiment
4. Thermal stress
5. Gate current: dominant charge transport mechanisms
6. Conclusions

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# Benefits of GaN for RF

Promising applications:



Benefits of GaN for RF:

- Wide bandwidth
- High power density
- Excellent energy efficiency
- Small volume

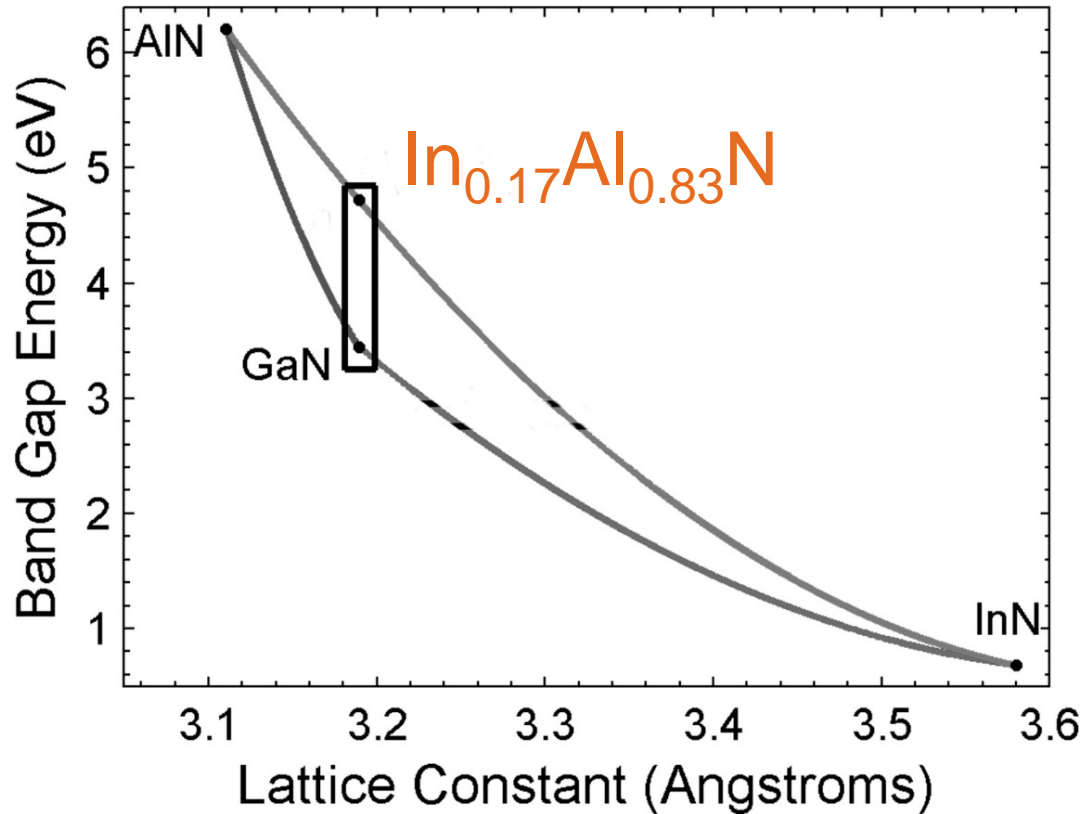
# InAlN as barrier material

	$\text{Al}_{0.2}\text{Ga}_{0.8}\text{N}/\text{GaN}$	$\text{In}_{0.17}\text{Al}_{0.83}\text{N}/\text{GaN}$
$\Delta P_0$ ( $\text{cm}^{-2}$ )	$6.5 \times 10^{12}$	$2.7 \times 10^{13}$
$P_{\text{piezo}}$ ( $\text{cm}^{-2}$ )	$5.3 \times 10^{12}$	0
$P_{\text{total}}$ ( $\text{cm}^{-2}$ )	$1.2 \times 10^{13}$	$2.7 \times 10^{13}$

[J. Kuzmik, EDL 2001]

- High spontaneous polarization in InAlN  $\rightarrow$  high 2DEG density
- InAlN thickness scaling  $\rightarrow$  gate length scaling  
 $\rightarrow$  W- and V-band applications

# InAlN as barrier material



[M. A. Laurent, JAP 2014]

$\text{In}_{0.17}\text{Al}_{0.83}\text{N}$  lattice matched to GaN

→ Potentially better reliability!

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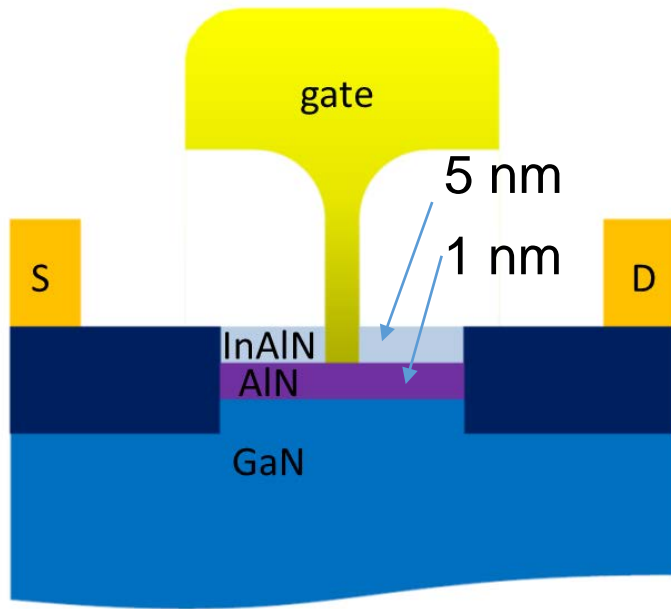
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# Devices (E-mode)

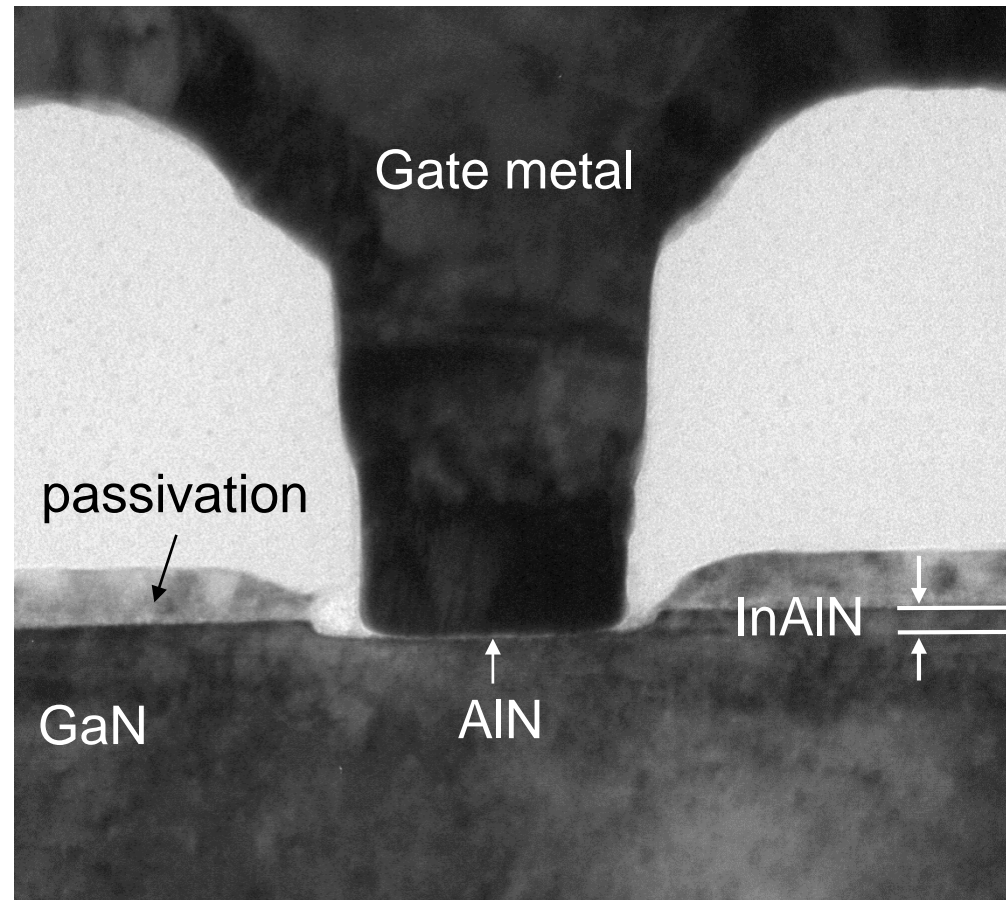
InAlN/AlN/GaN HEMTs:

- E-mode
- $L_g = 40 \text{ nm}$
- $L_{gs} = L_{gd} = 1 \text{ }\mu\text{m}$
- W-band



[Saunier, CSICS 2014]

BFTEM of virgin device



Thermal models available

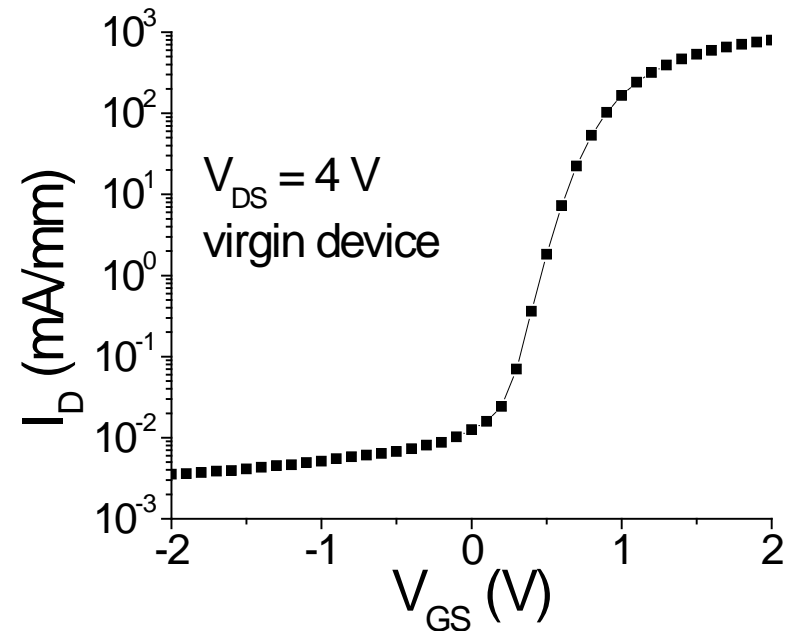
# FOMs for benign characterization, detrapping methodology

## FOMs:

- $I_{Dmax}$  :  $I_D$  at  $V_{GS} = 2 \text{ V}$  and  $V_{DS} = 4 \text{ V}$
- $V_{Tsat}$  :  $V_{GS}$  extrapolated from  $I_D$  at peak  $g_m$  point at  $V_{DS} = 4 \text{ V}$
- $I_{Goff}$  :  $I_G$  at  $V_{GS} = -2 \text{ V}$ ,  $V_{DS} = 0.1 \text{ V}$
- $I_{Doff}$  :  $I_D$  at  $V_{GS} = -2 \text{ V}$ ,  $V_{DS} = 0.1 \text{ V}$
- $R_D$  : at  $I_G = 20 \text{ mA/mm}$
- $R_S$  : at  $I_G = 20 \text{ mA/mm}$

## Detrapping & initialization:

- 100 °C bake for 1 hour



# Impact of characterization and detrapping

Impact of 200 successive characterizations

	$\Delta I_{Dmax}/I_{Dmax}(0)$ [%]	$\Delta V_{Tlin}$ [mV]	$R_D/R_D(0)$	$R_S/R_S(0)$
After initialization	0	0	1	1
After 200 characterizations	0.70	-23.4	0.95	0.89
After detrapping	0	2.3	1.01	1

Nearly complete recovery after thermal detrapping step  
→ characterization suite is benign and detrapping step is effective

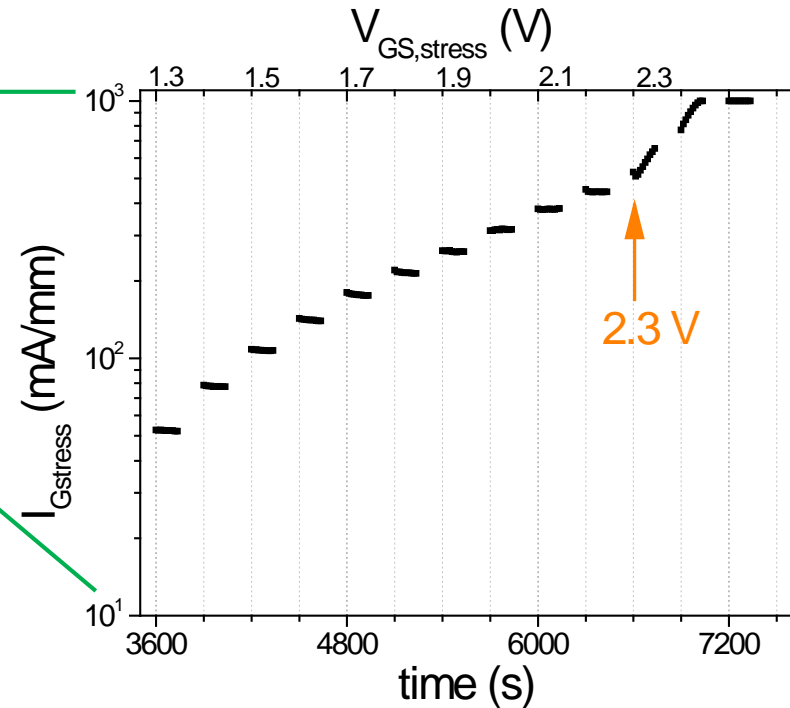
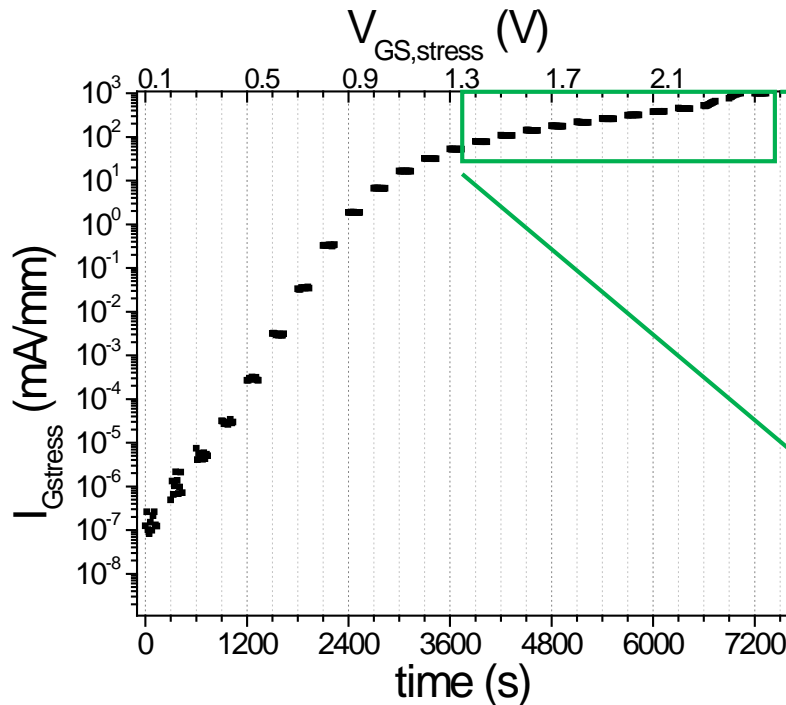
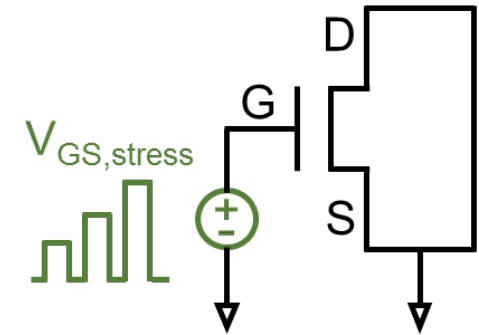
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# RT Positive- $V_G$ step-stress-recovery experiment

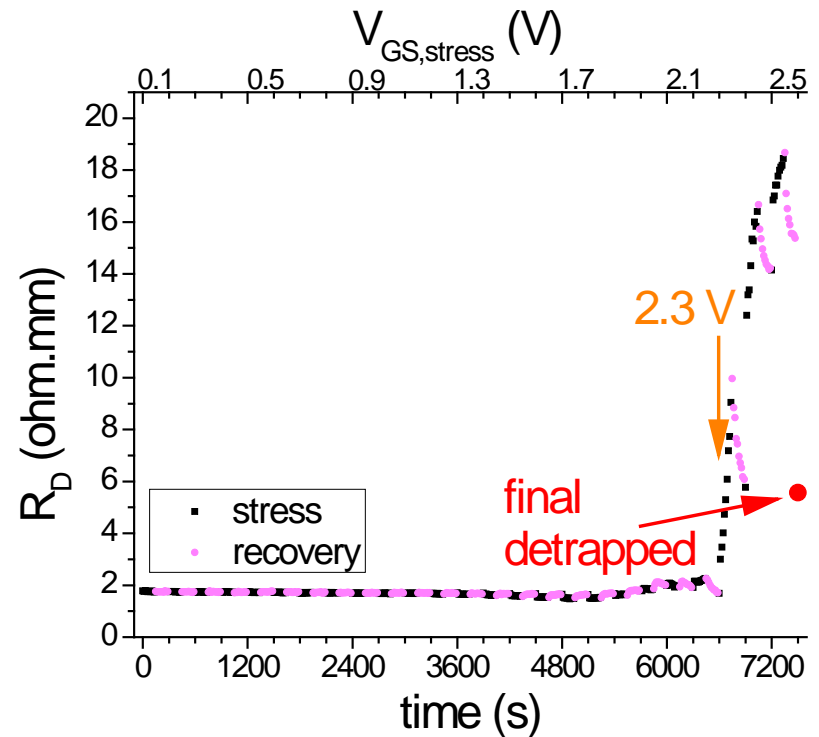
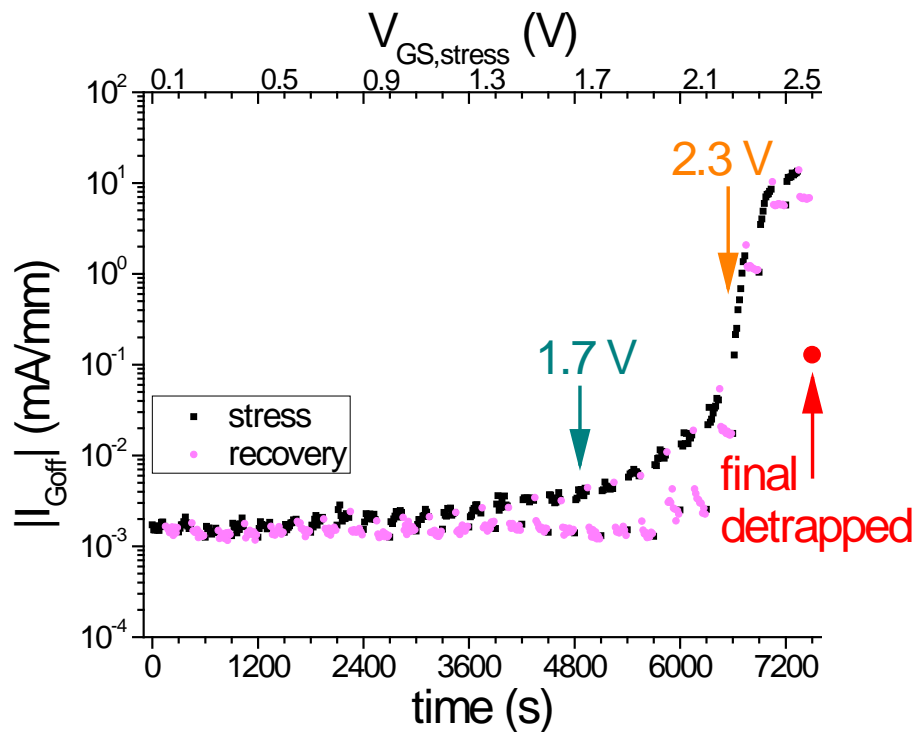
Stress conditions:

- $V_{GS, stress} = 0.1 - 2.5$  V in 0.1 V steps,  $V_{DS} = 0$  V, RT
- Recovery:  $V_{DS} = V_{GS} = 0$  V
- Stress time = recovery time = 150 s
- Characterization every 15 s



- $I_{Gstress} \uparrow$  at constant  $V_{GS, stress}$  for  $V_{GS, stress} \geq 2.3$  V

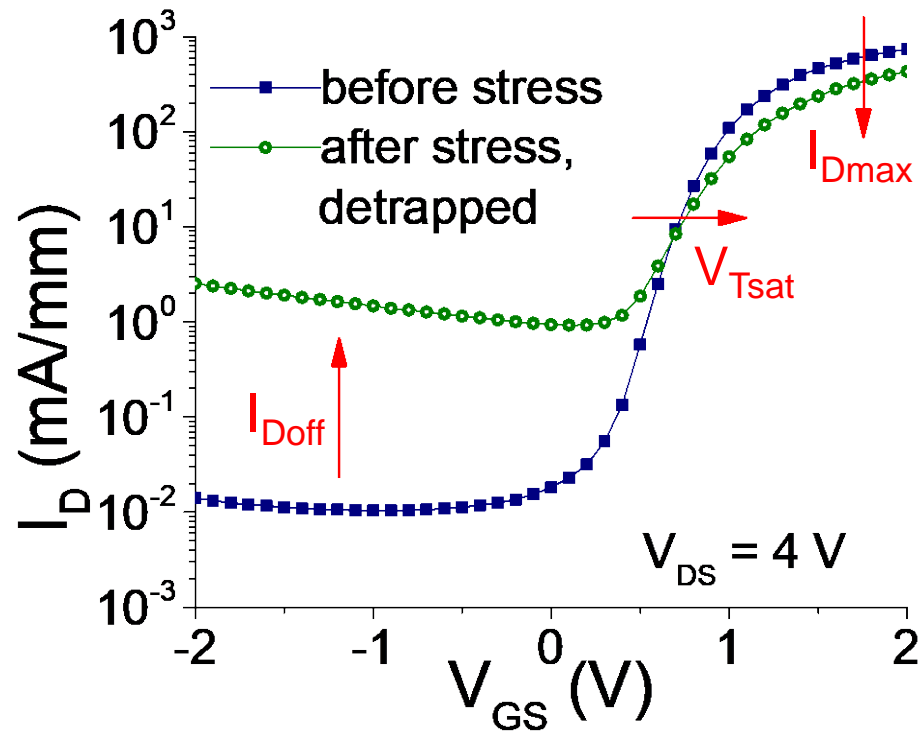
# Time evolution of $I_{\text{Goff}}$ and $R_{\text{D}}$



Two mechanisms:

- From  $V_{\text{GS, stress}} = 1.7$  V:  $I_{\text{Goff}} \uparrow$ , trapping  $\uparrow \rightarrow$  mechanism 1: new defects generated in AlN
  - From  $V_{\text{GS, stress}} = 2.3$  V:
    - $I_{\text{Goff}} \uparrow \uparrow$
    - $R_{\text{D}}$  and  $R_{\text{S}} \uparrow \uparrow$
- } Mechanisms 2 ?

# Before and after stress: permanent degradation

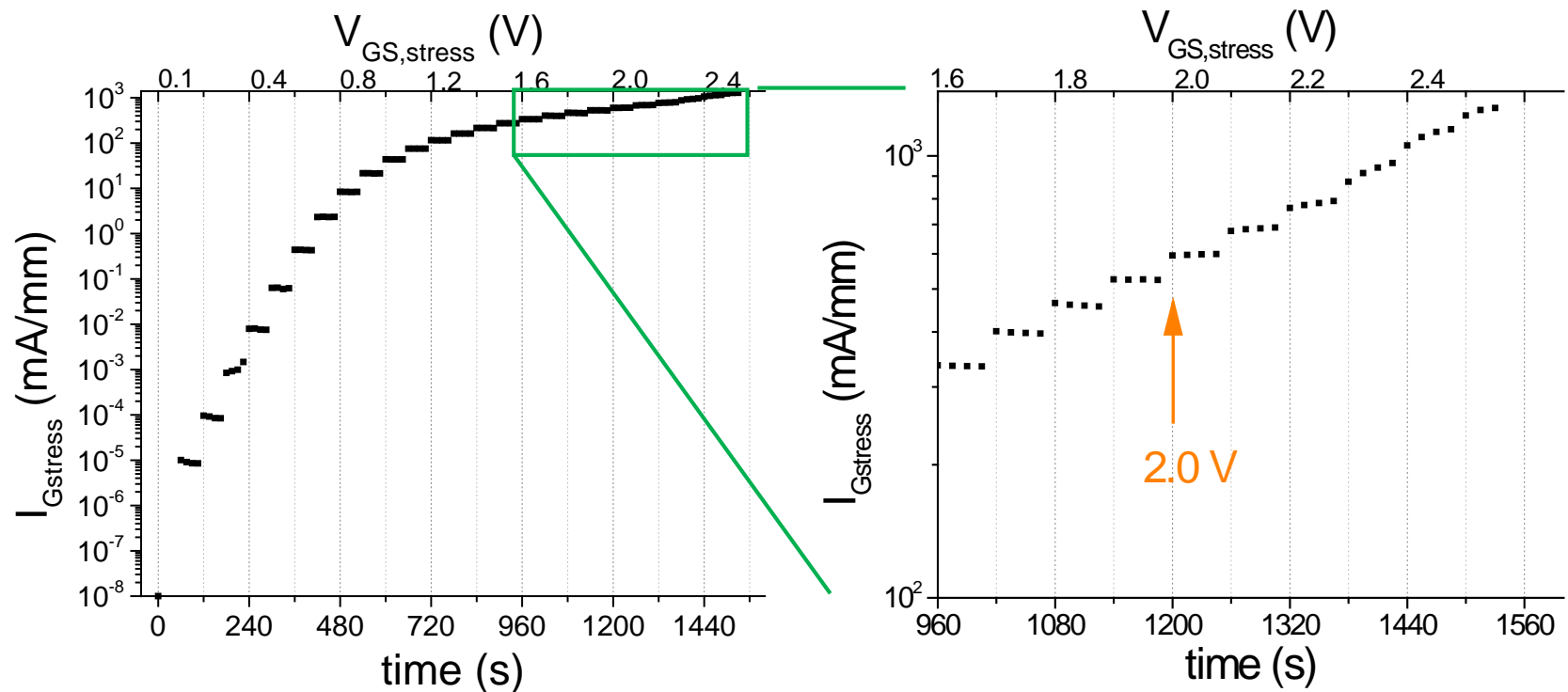


- $I_{Doff} \uparrow\uparrow$
  - $I_{Dmax} \downarrow$
  - $\Delta V_{Tsat} > 0$
- } Mechanisms 2: consistent with gate sinking

# High T Positive- $V_G$ step-stress experiment

Stress conditions:

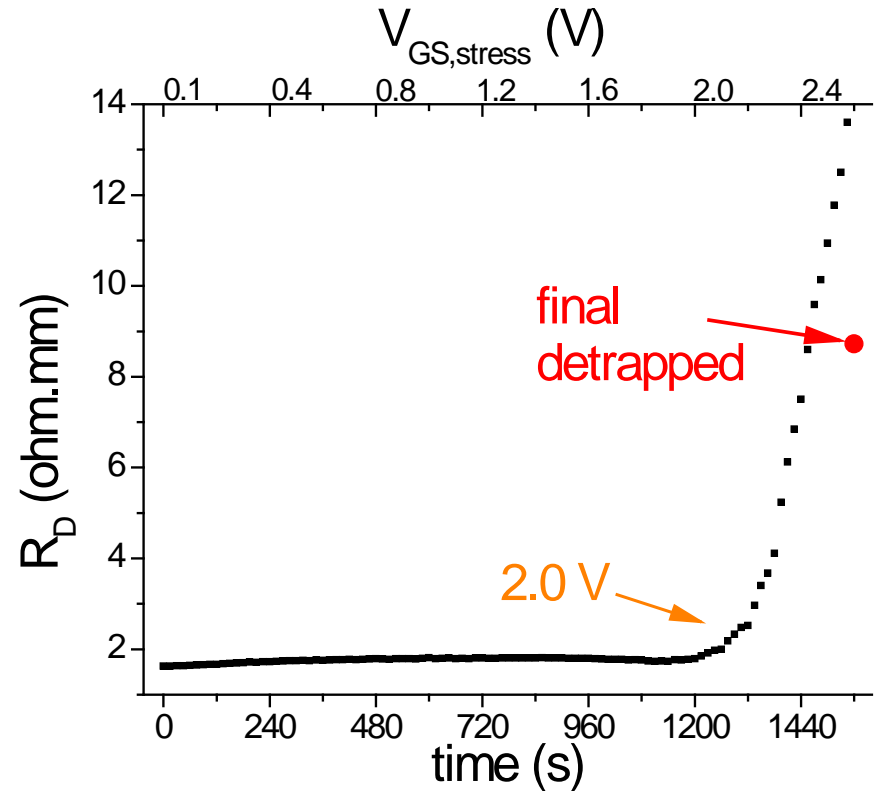
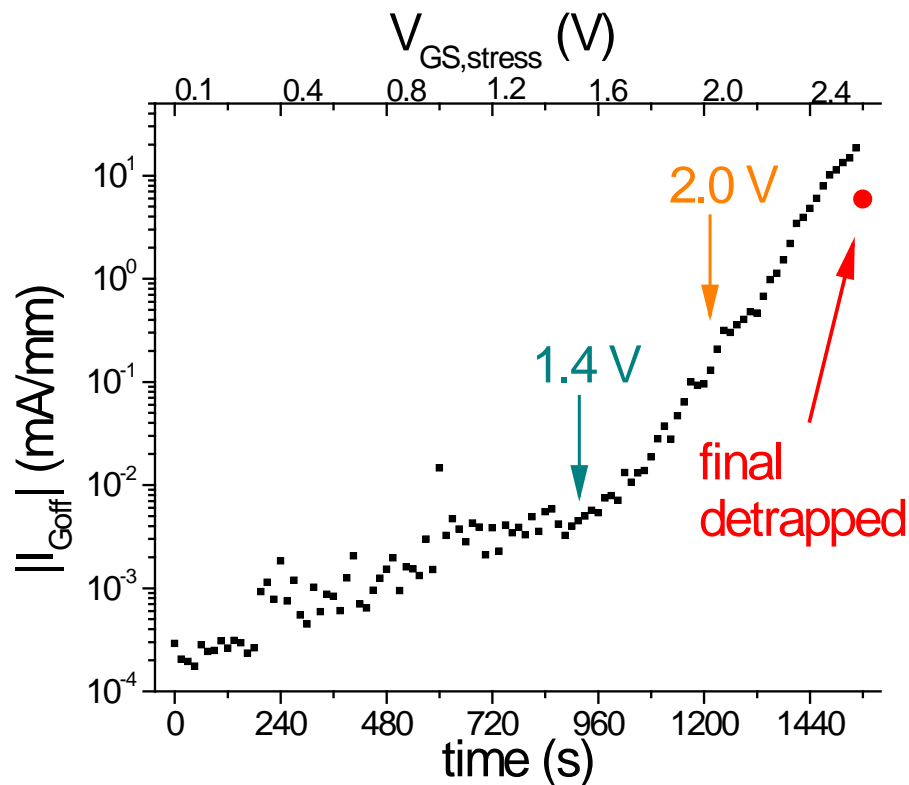
- $V_{GS, stress} = 0.1 - 2.5$  V in 0.1 V steps,  $V_{DS} = 0$  V,  $T_{stress} = 150$  °C
- Stress time = 60 s
- Characterization every 15 s



- $I_{Gstress} \uparrow$  at constant  $V_{GS, stress}$  for  $V_{GS, stress} \geq 2.0$  V



# High T Positive- $V_G$ step-stress experiment



- From  $V_{GS, stress} = 1.4$  V:  $I_{Goff} \uparrow$
- From  $V_{GS, stress} = 2.0$  V:  $I_{Goff}$ ,  $R_D$ , and  $R_S \uparrow \uparrow$
- Lower threshold for degradation than at RT  $\rightarrow$  Both mechanisms thermally enhanced

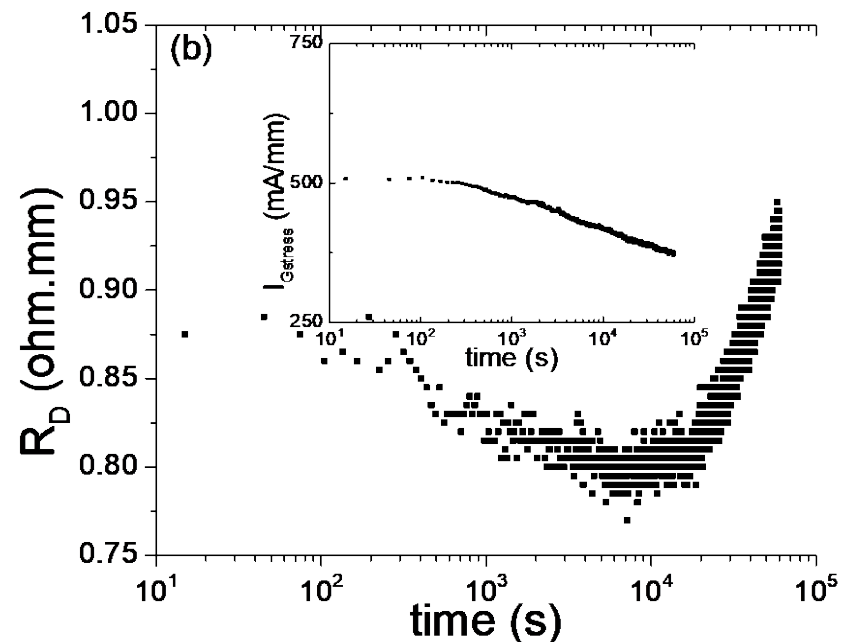
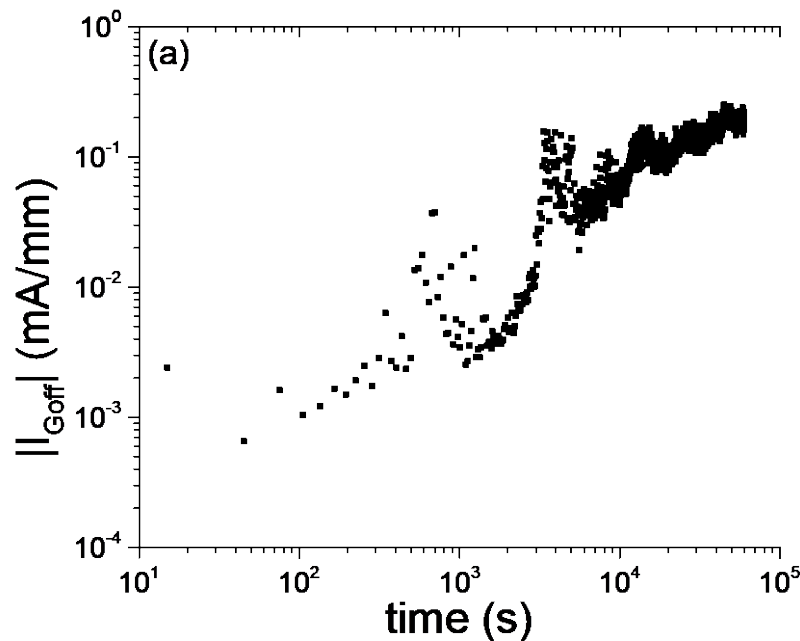
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# RT Constant- $V_G$ stress experiment

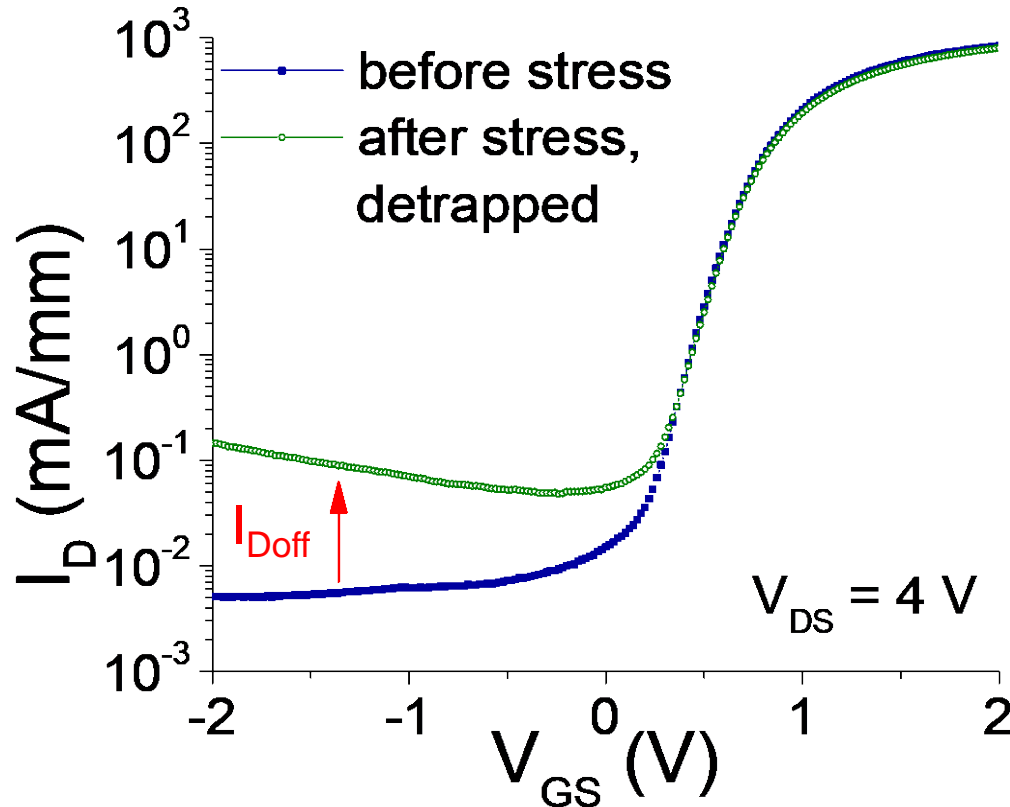
Stress conditions:

- $V_{GS, \text{stress}} = 2 \text{ V}$ ,  $V_{DS} = 0 \text{ V}$ , RT
- Characterization every 15 s



- $I_{\text{Goff}}$  becomes noisy at  $t_{\text{stress}} \sim 500 \text{ s}$ ; increases afterwards  
→ consistent with trap generation in AlN layer (mechanism 1)
- $R_D$  changes little throughout experiment
- $I_{\text{Gstress}}$  keeps decreasing → no Schottky barrier degradation

# Before and after stress: permanent degradation



- $I_{Doff} \uparrow\uparrow$
- No significant  $I_{Dmax}$  degradation
- No significant subthreshold characteristics degradation

# Summary so far

Two degradation mechanisms identified:

## 1. Under mild gate stress:

- Observation:  $I_{\text{Goff}} \uparrow$ , trapping  $\uparrow$ , thermally enhanced
- Proposed mechanism 1: high electric field induced defect generation in AlN interlayer

## 2. Under harsh gate stress:

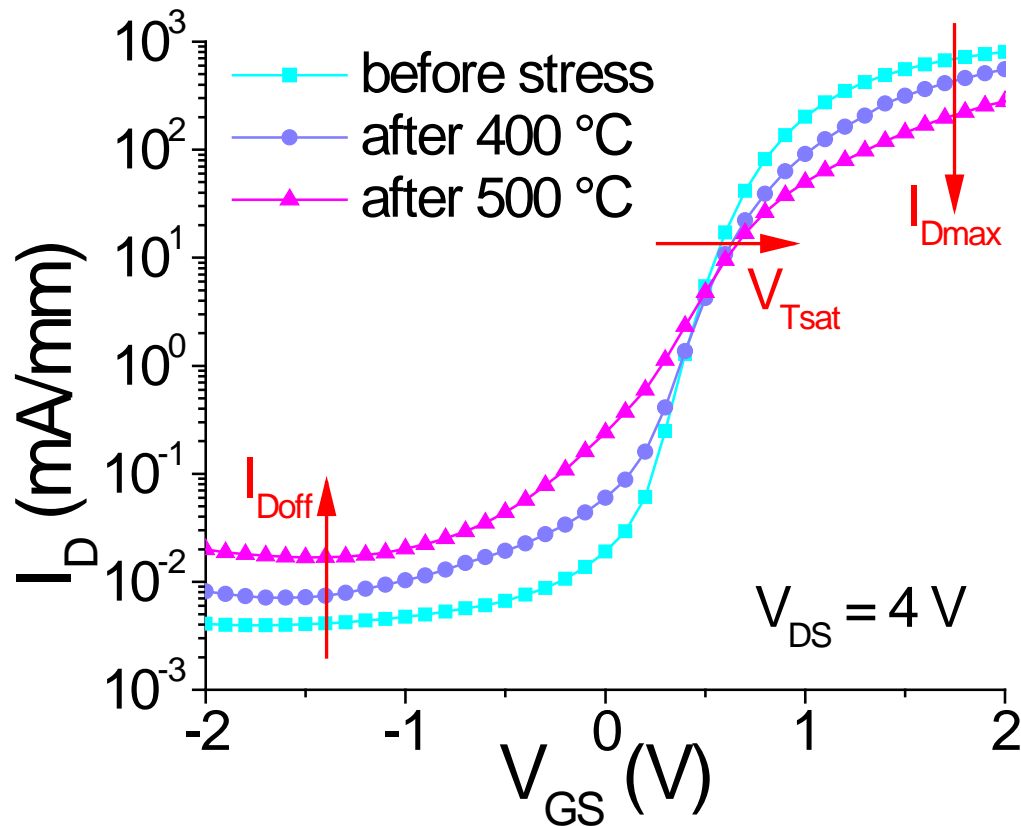
- Observation:  $I_{\text{Goff}} \uparrow$ ,  $R_D$  and  $R_S \uparrow$ ,  $\Delta V_T > 0$ ,  $I_{D\text{max}} \downarrow$ , thermally enhance
- Proposed mechanism 2: self-heating induced Schottky gate degradation, or gate-sinking

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# Thermal stress experiment

Impact of thermal stress: 1 min. RTA in N<sub>2</sub>



Permanent degradation:

- Prominent  $I_{Dmax} \downarrow$  with T
- Positive  $V_{Tsat}$  shift
- $I_{Doff} \uparrow$

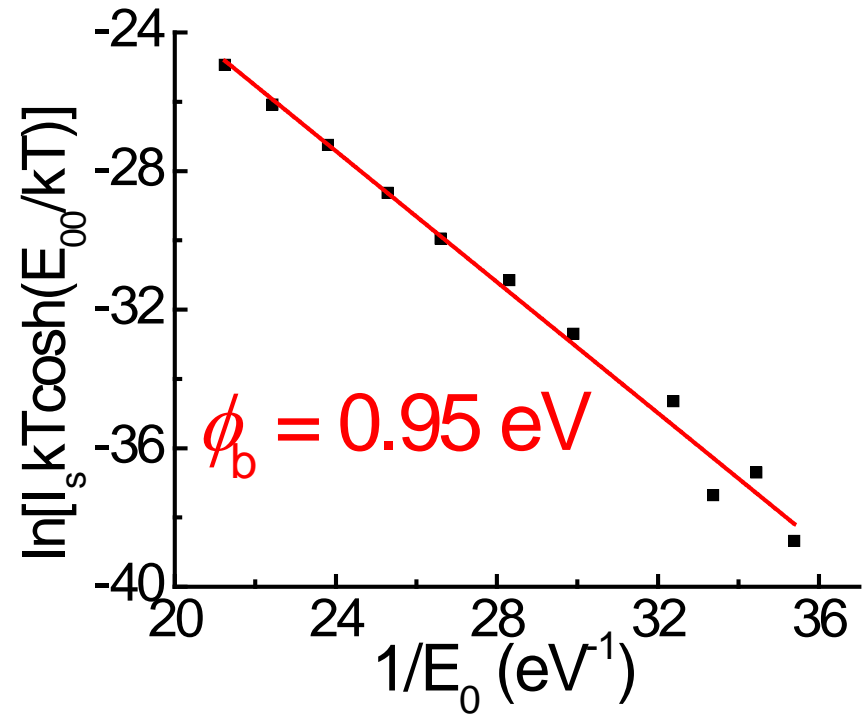
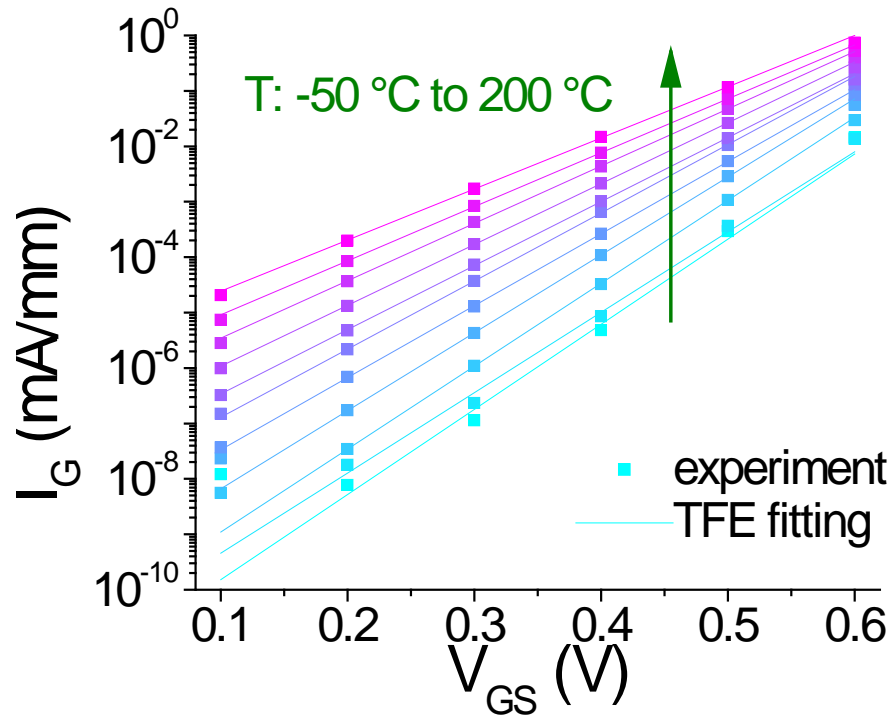
Same signature as that of degradation mechanism 2  
→ consistent with gate sinking

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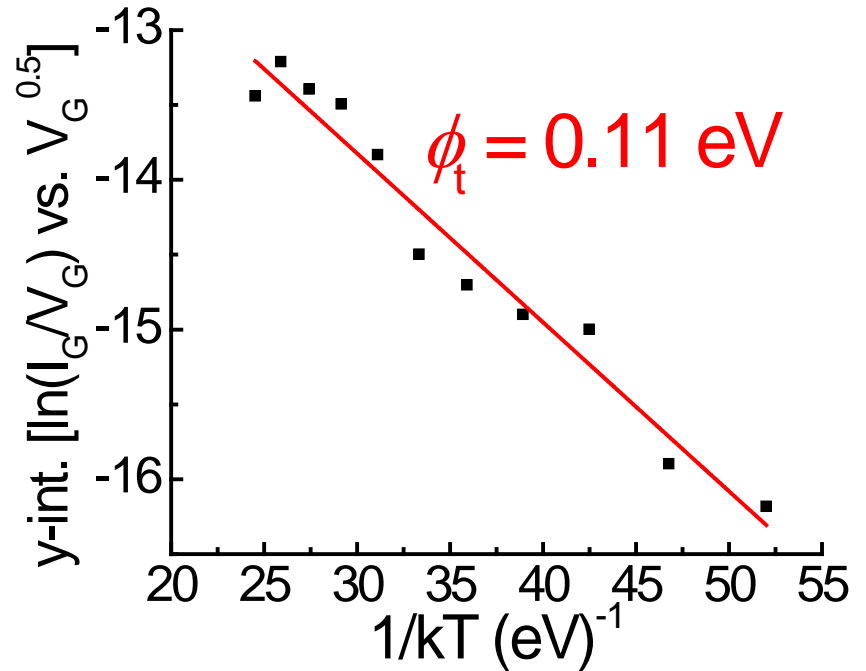
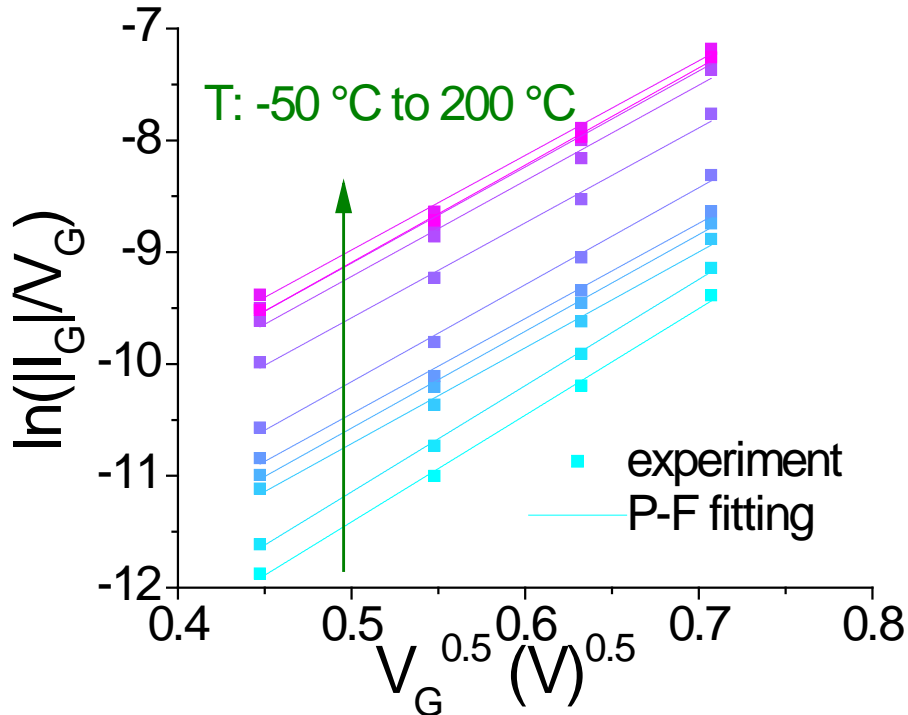
# Virgin device: Thermionic Field Emission fitting



- T dependence well explained by Thermionic Field Emission (TFE) theory
- Extracted effective Schottky barrier height ( $\phi_b$ ): 0.95 eV

# After harsh gate stress: Poole-Frenkel Emission fitting

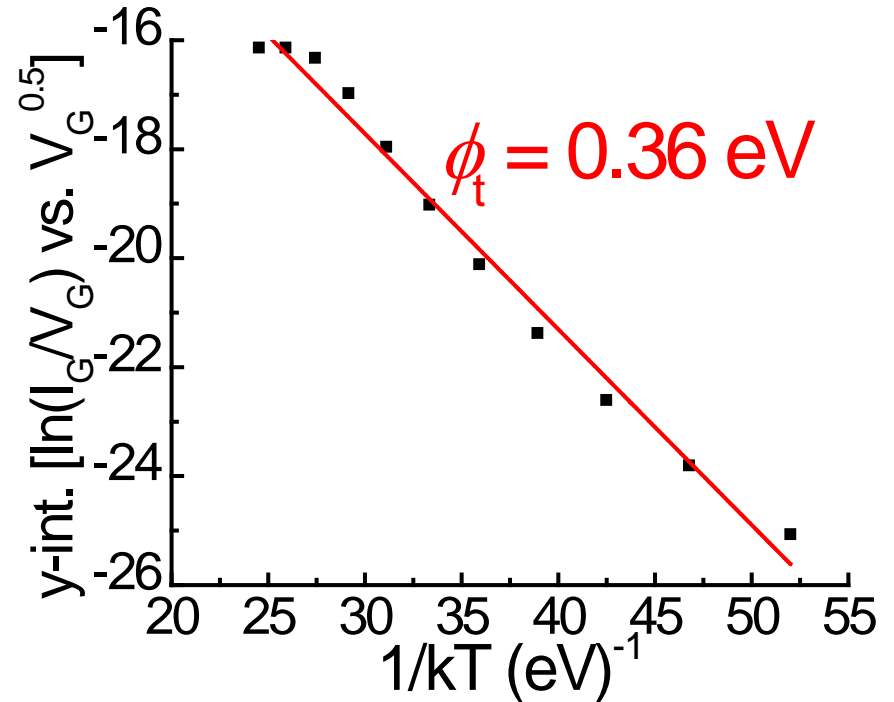
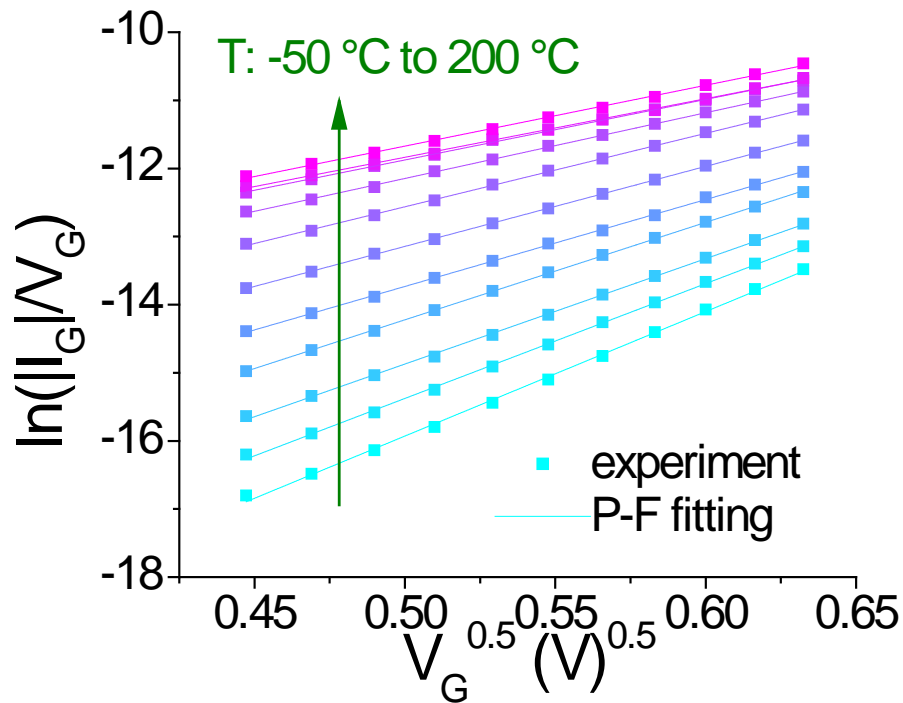
Stress conditions:  $V_{GS, stress} = 0 - 2.5$  V in 0.1 V steps,  $V_{DS} = 0$  V



- T dependence well explained by Poole-Frenkel Emission (P-F) theory
- Extracted effective trap energy level ( $\phi_t$ ): 0.11 eV

# After mild gate stress: Poole-Frenkel Emission fitting

Stress conditions:  $V_{GS, stress} = 2 \text{ V}$ ,  $V_{DS} = 0 \text{ V}$



- T dependence well explained by Poole-Frenkel Emission (P-F) theory
- Extracted effective trap energy level ( $\phi_t$ ): 0.36 eV
- Close to donor level of N vacancy in AlN of 0.5 eV [T. L. Tansley, PRB 1992]

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# Conclusions

Identified two degradation mechanisms:

## 1. Under mild gate stress:

- Observation:  $I_{\text{Goff}} \uparrow$ , trapping  $\uparrow$ , thermally enhanced
- Proposed mechanism 1: high electric field induced defect generation in AlN interlayer

## 2. Under harsh gate stress:

- Observation:  $I_{\text{Goff}} \uparrow$ ,  $R_D$  and  $R_S \uparrow$ ,  $\Delta V_T > 0$ ,  $I_{\text{Dmax}} \downarrow$ , thermally enhanced
- Proposed mechanism 2: self-heating induced Schottky gate degradation, or gate-sinking

Transport model for  $I_G$  in low forward regime:

- Virgin device: TFE with  $\phi_b = 0.95$  eV
- After degradation mechanism 1: P-F with  $\phi_t = 0.36$  eV
- After degradation mechanism 2: P-F with  $\phi_t = 0.11$  eV